

Surface Integrity Characterization Based on Time-Delay of the Magnetic Barkhausen Noise Voltage Signal

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The captured magnetic Barkhausen noise (BN) signal is composed of a series of voltage impulses changes produced by movements of the magnetic domains. In most cases the captured voltage signals cannot be directly related to individual parameters to assess the material state, i.e. the properties, of the surface layer. For further efficient analysis of the voltage signals, an appropriate method for signal processing should be chosen in order to use the characteristic value of the voltage signal.

Our method of processing represents the time delay of a BN voltage signal, i.e. the maximum voltage value of the signal, with reference to the sine wave of the magnetizing current. Measurement of the time delays of the voltage signals was carried out under optimum magnetizing conditions, which gave us the greatest number of voltage impulses in the signal. These are influenced by the turns and orientation of the magnetic domains.

The aim of this research was to compare the voltage-signal time delay obtained with quenched and tempered specimens at two neighbouring temperatures. The temperature differences ranged from $\Delta T_1 = 10$ °C, up to $\Delta T_2 = 25$ °C. These small temperature differences in the tempering-temperature produced small differences in the micro hardness of the individual specimens and found out whether there were significant differences between micro hardnesses. The assessment of the reliability of the prediction of micro hardness was carried out using the Student's t-test.

Keywords: micro-magnetic method, Barkhausen noise, time delay of the voltage signal, micro hardness, Student's t-test, reliability

0 INTRODUCTION

Many authors have reported different methods of measuring and processing the captured Barkhausen noise (BN) voltage signals in their research.

Jiles and Suominen [1] found that micro-hardness and residual stresses were determined from the captured voltage signal of Barkhausen noise according to the measured depth. They found that at the same analyzing frequency and given specific electric conductivity and relative permeability of the material, a smaller depth of the micro magnetic change was obtained and vice versa. Wojtas and Suominen [2] and [3] stated that the X-ray measurement method is a reliable method for residual stress characterization, but it is slow and appropriate only for laboratory conditions. On the other hand, the magnetic Barkhausen noise (MBN) technique is fast and based on calibration curves for the determination of residual stresses. Savaş and Gür [4] studied non-destructive evaluation of surface residual stresses in shot peened steel components using the magnetic Barkhausen noise method. For this purpose, various sets of steel specimens were prepared by a controlled shot peening process with different intensities, impact angles and coverage values. The measurements showed that a clear relationship exists between surface residual stresses and the Barkhausen noise signals. Grum and Žerovnik [5] also assessed the residual stresses in

steel 1C40 that had been hardened and tempered. The assessment of the efficiency of the determination of residual stresses by the micro-magnetic method based on the BN was carried out by means of measurement of residual stresses using the relaxation method. Comparative through-thickness measurements showed that the micro-magnetic method was suitable for the determination of residual stresses since it gives a real-time variation of residual stresses in a thin surface layer of the material concerned in a very short time. Žerovnik et al. [6] assessed the microstructure, hardness, and residual stresses on induction surface-hardened specimens.

The most commonly chosen feature of the BN is the square mean value of the voltage signal V_{RMS}^2 . Desvaux et al. [7] studied ultrasonic shot peening, which allows for control of the introduction of residual stresses on the raceways and increases the fatigue resistance of the bearings. The level of residual stresses that are introduced can be verified by BN at any point on the bearing raceway surface and subsurface and guaranteed to 100% on a complete manufacturing lot. Kikuchi et al. [8] determined the magnetic BN energy and the peak in the V_{RMS}^2 voltage signal reflects the number of pinning sites for domain wall motion. These therefore rise sharply in an initial stage of cold rolling and almost saturate at a higher reduction ratio. On the contrary, the coercive force and the magnetizing current at the peak in the V_{RMS}^2

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voltage increase monotonically due to the increase in dislocation density below 10% and the formation of the cell structure. Žerovnik and Grum [9] discovered that the microstructure and micro hardness have a significant effect on the relative permeability μ_r and measurement depth z . The results showed that the damping of the micro magnetic parameter with depth reaches its maximum in the soft specimen, and vice versa. Lo et al. [10] studied the effects of microstructural variations with depth on the Barkhausen effect (BE) signals in surface-modified ferrous materials through measurements and simulations based on a hysteretic-stochastic model. Theoretical analysis showed that the model parameters, which describe the domain-wall pinning strength and the range of interaction of a domain wall with pinning sites, are related to each other. Pepelnjak and Barišič [11] have measured the strains occurring during sheet material plastic deformation using an optical measurement system. The strains were measured up to the tearing limit in order to determine the forming limit diagram using the Marciniak testing method. Due to the plane strain problem and fixing the focus distance between the observed sheet metal surface and the lens, only one charge-coupled device (CCD) camera was used. Sablik et al. [12] presented one model of the effect of plastic deformation on magnetoacoustic emission (MAE), where one must first treat the non-180deg domain wall motion. In this paper, they take the Alessandro-Beatrice-Bertotti-Montorsi (ABBM) model and modify it to treat non-180deg wall motion. They then insert a modified stress-dependent Jiles-Atherton model, which treats plastic deformation, into the modified ABBM model to treat MAE and MBN. An experimental study by Vashista and Paul [13] was undertaken to investigate the role of process parameters on the grindability of medium carbon steel in a high-speed grinding domain with particular emphasis on surface integrity. Surface residual stress on the ground specimens has been assessed using X-ray diffraction techniques and BN analysis. High-speed grinding with cubic boron nitride (CBN) wheels, unlike conventional grinding, provided compressive residual stress throughout the experimental domain. This can be attributed to the desirable temperature control since the single layer CBN wheel had higher thermal conductivity than conventional wheels and grinding fluid took away a substantial part of the grinding heat flux. Micromagnetic or BN parameters correlated linearly with the residual stress, indicating its applicability in assessing the surface integrity of high-speed ground steel. Koomatsubara et al. [14] captured the BN of soft

ferrites and silicon steels. The experimental data are discussed in terms of the relevant quantities: saturation flux density (B_s), conductivity σ , cross-sectional area of samples (SO) and grain diameter (d). At low field velocities $\text{mod}(H)$, magnetization processes are dominated by independent displacements of 180 degrees walls. The peak noise power (P_m) in the vicinity of the coercive field is proportional to the field velocity $\text{mod}(H)$. These two simple results reflect nonoverlapping Barkhausen pulses. Krause et al. [15] compares the anisotropy of surface MBN and the pulse height distributions between two samples of oriented 3% Si-Fe steel laminate at equivalent flux densities. Differences between the two samples were observed in the anisotropic behavior of the MBN-Energy and pulse height distributions for flux density amplitudes between 1.3 and 1.6 T, the typical operating range of transformer laminates. These were attributed to differences in the number and structure of the 180° domain walls. Schneider et al. [16] studied the surface acoustic waves as a means of evaluating surface-hardened steels. A generalized dispersion curve was calculated using the theory of the surface acoustic wave to deduce the hardening depth from the velocity. Using this methodology, it was found that the hardening depth could be determined with an error not higher than 15%.

1 EXPERIMENTAL SET-UP

For these investigations an experimental setup was arranged to capture voltage signals of the MBN. It consisted of a magnetisation unit, a sensor for capturing voltage signals, a signal amplifier with a relevant band-pass filter, and a computer-aided unit for determination of the microstructure or micro hardness and residual stresses. Fig. 1 shows a block scheme of the experimental setup for micromagnetic testing based on the BN.

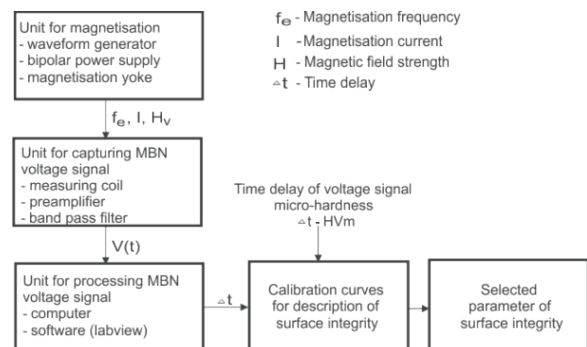


Fig. 1. Experimental setup for capturing the voltage signal of the magnetic Barkhausen noise

In the analysis of the BN at the chosen depth, it was important to filter the BN voltage signals. A frequency filter provided a signal of the voltage induced in the measuring coil sensing the changes during the process of specimen magnetization. Thus a Butterworth filter of the fourth order was applied in a series connection with a low-pass and high-pass filter of the second order. Thus, in the experimental setup, four different band-pass Butterworth filters of the fourth order pass signals in different frequency ranges.

2 THE ANALYZED MATERIAL AND SELECTION OF OPTIMAL MAGNETIZING PARAMETERS

In our research, we used 1C55 heat-treated carbon steel with a 0.58% carbon content, which is used in the production of various machine parts exposed to heavy loading.

Table 1. Chemical composition of steel 1C55 [wt.%]

C	Si	Mn	Cr	Ni	Cu	Al	Sn	Mo	V	Ti
0.58	0.26	0.63	0.10	0.15	0.28	0.02	0.013	0.03	0.01	0.002

For this purpose, ferromagnetic steel bands of 150×30×5 mm were subjected to three types of heat treatment:

- Soft-annealed specimens, annealed at a temperature of $T_{SA} = 690$ °C, with an average microhardness of 224 HV_{0.2}.
- Specimens quenched in oil at a temperature of $T_H = 850$ °C, with an average micro hardness of 803 HV_{0.2}.
- Specimens quenched in oil at a temperature of $T_H = 850$ °C, tempered to a temperature of $T_T = 300$ °C, with an average micro hardness of 511 HV_{0.2}.

After different heat treatments of steel 1C55 we obtained different structural changes and different profiles of micro hardness.

At the soft annealed temperatures $T_{SA} = 690$ °C, a fine pearlite-ferrite microstructure forms. The size and density of the coagulated cementite in the pearlite grains determine steel hardness after heat treatment (Fig. 2a). The specimens quenched in oil at a temperature of $T_H = 850$ °C show a microstructure of fine tetragonal martensite (Fig. 2b). With the specimens quenched in oil at a temperature of $T_H = 850$ °C and tempered to a temperature of $T_T = 300$ °C, martensite will disintegrate into a bainite microstructure. Thus, the bainite microstructure shows

very fine feathers of cementite in the ferritic matrix (Fig. 2c).

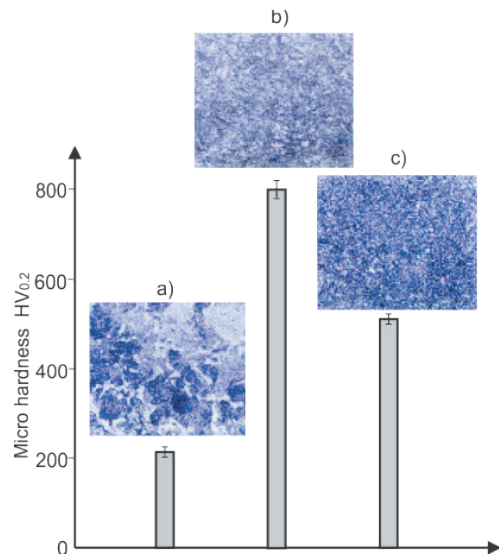


Fig. 2. The microstructure and micro hardness of individual heat-treated 1C55 steel specimens; a) soft annealed, b) quenched in oil, c) quenched and tempered

Before initiating these experiments, it was necessary to select the optimum magnetization parameters. Various magnetization parameters affect the shape of the signals captured and the number of voltage impulses in the signal. In the first phase, the number of voltage impulses in the voltage signal was determined as a function of the magnetizing frequency (f_c), and the magnetizing current (I). The experimental work accomplished hitherto indicates that voltage signals having a higher density of voltage impulses provide more useful data on the material.

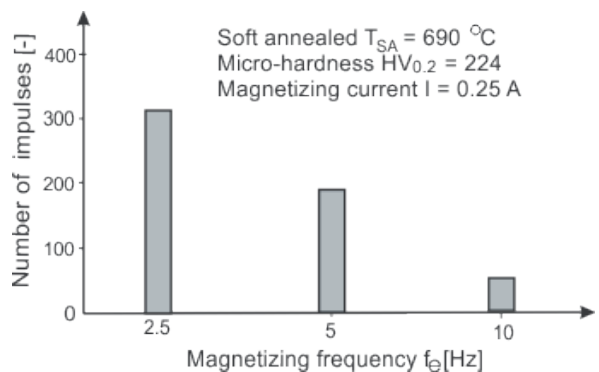


Fig. 3. Density of voltage impulses in the signal as a function of the magnetizing frequency and constant magnetizing current

The Barkhausen voltage signals were captured from soft annealed 1C55 carbon steel at a temperature

T_{SA} of 690 °C. Fig. 3 shows the number of voltage impulses in the signal as a function of the magnetizing frequency and a constant magnetizing current. Fig. 4 shows the number of voltage impulses in the signal as a function of the magnetizing current and a constant magnetizing frequency.

The diagrams in Figs. 3 and 4 indicate that the density of impulses in the voltage signal is reduced if the magnetizing frequency and the magnetizing current increase and vice versa.

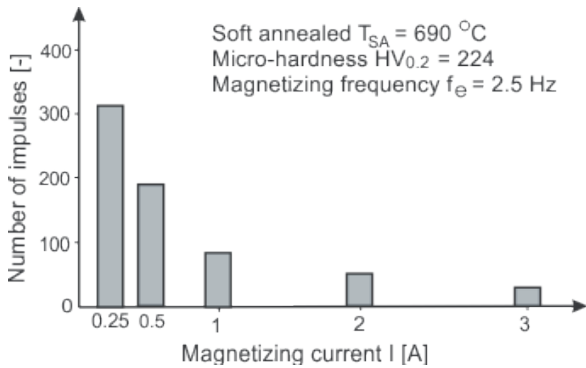


Fig. 4. Density of voltage impulses in the signal as a function of the magnetizing current and constant magnetizing frequency

3 PROCESSING THE BN VOLTAGE SIGNAL ON THE BASIS OF TIME DELAYS

The captured BN voltage signal is composed of a series of voltage impulses. In the first phase the voltage impulses rise as they near the centre of the voltage signal and then decrease in the same manner toward zero. Changes in voltage impulses are caused by the movements of magnetic domains. The whole captured voltage signal of an MBN is unstationary. In limited parts of the voltage signal the stationary can be defined as stationary in the narrow or the wide part of the signal. In the central part of the captured voltage signal we can assume the stationary in wide range. This means that when evaluating the voltage signal it is sufficient that the first two central moments (arithmetic mean of the signal m_x^2 and signal variance σ^2) are time independent.

In most cases the captured voltage signals cannot be directly related to individual parameters to assess the state, i.e. properties, of the surface layer. For further efficient analysis of the voltage signals, an appropriate method for signal processing should be chosen in order to use the characteristic value of the voltage signal.

In the analysis of the captured voltage BN signals we noticed various time delays according to

the sinusoidal of the magnetizing current. In the first phase, we sketched out an envelope of the captured voltage signal of BN. The time delay was measured from the highest point of the envelope (which also represents the area with the highest voltage impulses) to the sinusoidal of magnetizing current, as shown in Fig. 5.

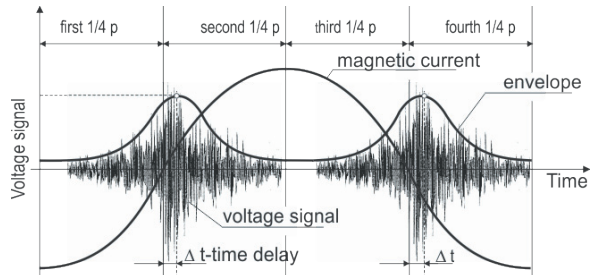


Fig. 5. Definition of the time delay of the BN voltage signal with reference to the sinusoidal of the magnetizing current

We found that different magnetizing parameters, as the current of the material, affected the size of the time delays. In the analysis of the magnetizing parameters we focused on the two most important ones, which were the magnetizing current I and the magnetizing frequency f_e . The current of the material is affected by heat treatment with consequent changes in microstructure and hardness, in the mechanical treatment with cutting, on the level of plastic deformation and so on.

The plot of the envelope of the voltage signal and the measurement of time delays from the sinusoidal of the magnetizing current were carried out with computer aid.

3.1 The Influence of Various Magnetizing Parameters on the Time Delays of BN Signals

Figs. 6 and 7 show the time delays of the voltage signals with different magnetization parameters. We need to change both the magnetizing current and the magnetizing frequency f_e . Fig. 6 shows the time delay of the signals captured with different magnetizing current intensities and a constant magnetizing frequency f_e of 2.5 Hz. The longest time delays are obtained with the lowest magnetizing current intensities so that with a current intensity I of 0.25 A, the time delay t equals 4.70 ms. The shortest time delay t , i.e. 0.80 ms, is obtained with a magnetizing current I of 3 A.

Fig. 7 shows the time delays of the signals as a function of different magnetizing frequencies and a constant magnetizing current. Higher magnetizing

frequencies produce shorter time delays so that with a frequency f_e of 10 Hz a time delay t of 1.40 ms is obtained whereas with a lower frequency f_e , i.e. 2.5 Hz, the time delay t of the signal is considerably longer, i.e. 4.70 ms.

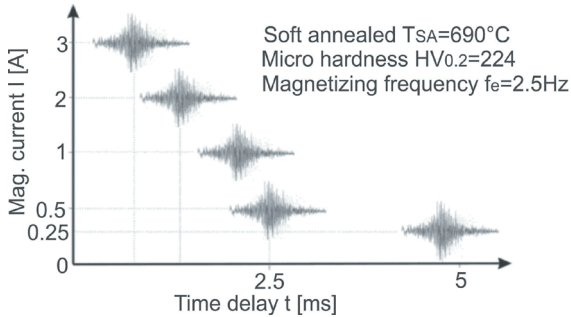


Fig. 6. Variation in the time delays of the maximum voltage values of BN signals as a function of different magnetizing current intensities and constant magnetizing frequency

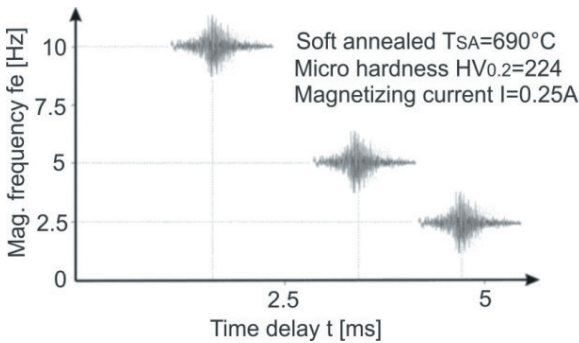


Fig. 7. Variation of time delays of the maximum amplitude values of BN signals as a function of different magnetizing frequencies and constant magnetizing current

3.2 The Influence of Heat Treatment on the Time Delay of Voltage Signals

We continued our research with the measurement of time delays of voltage signals. The measurements were performed on specimens that were heat treated with consequent change in microstructure and micro hardness (Fig. 2).

Measurement of the time delays of the voltage signals was carried out under optimum magnetization conditions. The optimum magnetization conditions are the ones giving the highest number of voltage impulses in the signal. They are influenced by the turns and orientation of the magnetic domains. Based on the given criteria related to the density of impulses in the BN voltage signal and the length of time delays of the signal, it can be stated that:

- the highest density of voltage impulses is obtained with a magnetizing frequency f of 2.5 Hz and a magnetizing current I of 0.25 A and
- the longest time delays t , i.e. 4.70 ms, are obtained with a magnetizing frequency f of 2.5 Hz and a magnetizing current I of 0.25 A (to a specimen in the soft annealed state).

Data on time delays of the voltage signals, including the calculated mean values and data on microhardness are given in Table 2. The data indicate that there are differences in the time delays of the individual specimens in different states. For the specimens with higher hardness, a longer time delay is obtained and vice versa.

Table 2. Mean values of the time delays of the voltage signals with data on microhardness for the individual heat treated specimens

Heat treatments	Micro hardness HV _{0.2}	Time delays (mean values) t [ms]
Quenched, $T_H = 850^\circ\text{C}$	803	13.1
Quenched and tempering $T_H = 850^\circ\text{C}$, $T_T = 300^\circ\text{C}$	511	9.3
Soft annealed, $T_{SA} = 690^\circ\text{C}$	224	4.7

Fig. 8 shows microhardness $HV_{0.2}$ of the specimens in three different states as a function of time delays of the voltage signals.

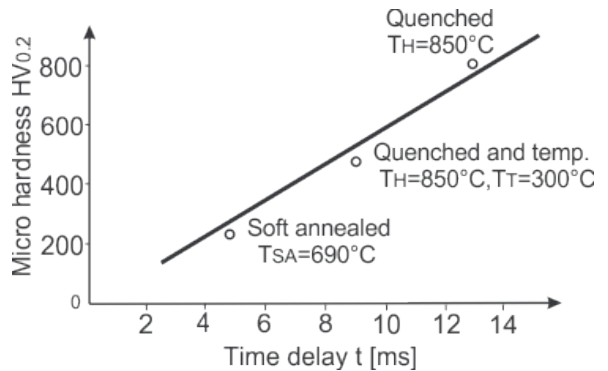


Fig. 8. Microhardness as a function of time delays from specimens of steel 1C55 in three different states

Results of the statistical analysis between the time delays of the voltage signals and the corresponding hardness are very favourable since the calculated correlation coefficient $r_{t/HV}$ is equal to 0.989.

4 DETERMINATION OF THE MICRO HARDNESS AND RELIABILITY ASSESSMENT

Determination of the micro hardness is based on a calibration curve. The calibration of the experimental system and elaboration of the calibration curves were

followed by measurements, and later a classification of the specimens with an unknown state. The calibration procedure and determination of micro hardness by means of the calibration curve is shown in Fig. 9.

The left part of Fig. 9 shows the procedure for elaborating the calibration curves using etalons with a known micro hardness, whereas the right part shows the procedure for testing an unknown specimen, i.e. determination of the characteristics of the surface-hardened layers.

The time delay was measured from the highest point of the envelope (which also represents the area with the highest voltage impulses) to the sinusoidal of the magnetizing current. The plot of the envelope of voltage signal and the measurement of time delays from sinusoidal of magnetizing current were carried out with computer aid. The data indicate that there are differences in the time delays of the individual specimens in different states. For the specimens with higher hardness, a longer time delay is obtained and vice versa. The following findings can be stated:

- When measuring specimens with an unknown micro hardness or unknown specimens, a suitable calibration curve determined on the basis of a preliminary measured micro hardness after heat treatment should be selected;
- A suitable calibration curve having been selected, measurement is carried out on the specimens with an unknown micro hardness. The time delays are then measured. By means of the calibration curve, the corresponding micro hardness value is determined.

The analysis of the results obtained made it possible to determine the reliability of the assessment of micro hardness of a given specimen based on the time delay of the voltage signal captured. The assessment of the reliability of predicting micro hardness was carried out using the Student's *t*-test.

Micro hardness measurement reliability analysis was conducted on the hardened and tempered specimens. For each high-tempering temperature three specimens were available. Three measurements were taken for each specimen, which means that nine measurements at the same tempering temperature were performed.

The aim of the research was to compare the voltage-signal time delay obtained with the quenched and tempered specimens at two neighbouring temperatures. The temperature differences ranged from $\Delta T_1 = 10\text{ }^\circ\text{C}$, up to $\Delta T_2 = 25\text{ }^\circ\text{C}$. The second aim of the research was to distinguish small differences in the micro hardness of the two neighbouring specimens

and to find out whether there were significant differences.

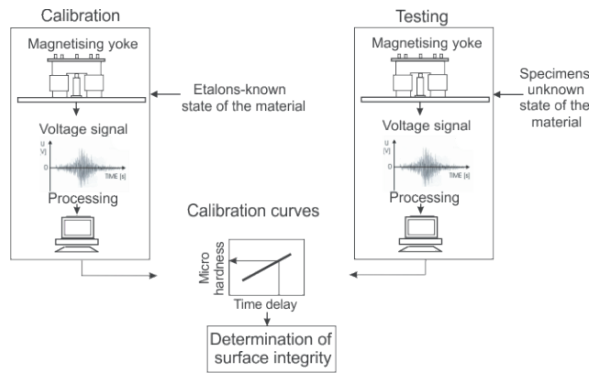


Fig. 9. Determination of micro hardness by means of calibration curves

In the study the quenched and tempered specimens tempered from $T_{T1} = 300\text{ }^\circ\text{C}$ to $T_{T5} = 400\text{ }^\circ\text{C}$ were used. Four different tempering temperatures rising gradually by $25\text{ }^\circ\text{C}$ were chosen.

Fig. 10 shows envelopes of the voltage signals captured from the samples in different states. The first envelope with a time delay $t = 4.85\text{ ms}$ belongs to the quenched specimen in oil from $T_H = 850\text{ }^\circ\text{C}$ and with the tempering temperature $T_T = 400\text{ }^\circ\text{C}$. The last envelope with a time delay $t = 9.30\text{ ms}$ belongs to the quenched specimen in oil from $T_H = 850\text{ }^\circ\text{C}$ and with the tempering temperature $T_T = 300\text{ }^\circ\text{C}$.

The harder specimens show a lower and flatter envelope with a longer time delay with regard to the sinusoidal of the magnetizing current and vice versa.

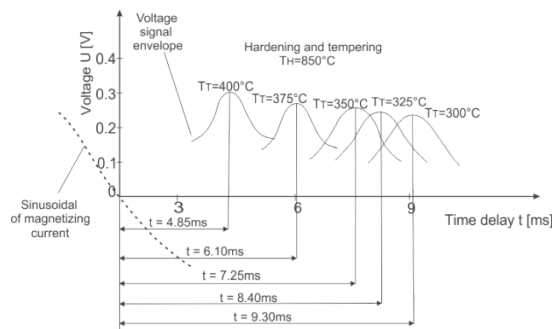


Fig. 10. Voltage signal envelopes for quenched and tempered specimens at different temperature

In Table 3 the comparative results of the micro hardness significance of the neighbouring temperatures are given for four different neighbouring tempering-temperature pairs only. A confidence limit was determined and hypotheses were chosen on the basis of the calculation of significance for the

Table 3. Test of micro hardness significance for different neighbouring tempering-temperature pairs ($\Delta T_2 = 25 \text{ }^\circ\text{C}$)

Neighbouring tempering temperature	Micro hardness from calibration curves (time delay)	Significance $a = 2P$	Limit of confidence P	Results
$(T_{T1} = 300 \text{ }^\circ\text{C} \text{ to } T_{T2} = 325 \text{ }^\circ\text{C})$	$(H_1 = 511 / H_2 = 482)$	0.000924	0.05	$(H_1 \neq H_2)$
$(T_{T2} = 325 \text{ }^\circ\text{C} \text{ to } T_{T3} = 350 \text{ }^\circ\text{C})$	$(H_2 = 482 / H_3 = 464)$	0.000382	0.05	$(H_2 \neq H_3)$
$(T_{T3} = 350 \text{ }^\circ\text{C} \text{ to } T_{T4} = 375 \text{ }^\circ\text{C})$	$(H_3 = 464 / H_4 = 439)$	0.000049	0.05	$(H_3 \neq H_4)$
$(T_{T4} = 375 \text{ }^\circ\text{C} \text{ to } T_{T5} = 400 \text{ }^\circ\text{C})$	$(H_4 = 439 / H_5 = 415)$	0.000013	0.05	$(H_4 \neq H_5)$

Table 4. Comparative results of micro hardness significance for neighbouring tempering-temperature pairs ($\Delta T_1 = 10 \text{ }^\circ\text{C}$)

Neighbouring tempering temperature	Micro hardness from calibration curves (time delay)	Significance $a = 2P$	Limit of confidence P	Results
$H(T_{T1} = 300 \text{ }^\circ\text{C} - T_{T2} = 310 \text{ }^\circ\text{C})$	$(H_1 = 511 / H_2 = 486)$	0.04876	0.05	$(H_1 \neq H_2)$
$H(T_{T2} = 310 \text{ }^\circ\text{C} - T_{T3} = 320 \text{ }^\circ\text{C})$	$(H_2 = 486 / H_3 = 478)$	0.23940	0.05	$(H_2 = H_3)$
$H(T_{T3} = 320 \text{ }^\circ\text{C} - T_{T4} = 330 \text{ }^\circ\text{C})$	$(H_3 = 478 / H_4 = 480)$	0.46178	0.05	$(H_3 = H_4)$
$H(T_{T4} = 330 \text{ }^\circ\text{C} - T_{T5} = 340 \text{ }^\circ\text{C})$	$(H_4 = 480 / H_5 = 462)$	0.04124	0.05	$(H_4 \neq H_5)$
$H(T_{T5} = 340 \text{ }^\circ\text{C} - T_{T6} = 350 \text{ }^\circ\text{C})$	$(H_5 = 462 / H_6 = 458)$	0.74187	0.05	$(H_5 = H_6)$
$H(T_{T6} = 350 \text{ }^\circ\text{C} - T_{T7} = 360 \text{ }^\circ\text{C})$	$(H_6 = 458 / H_7 = 451)$	0.25892	0.05	$(H_6 = H_7)$
$H(T_{T7} = 360 \text{ }^\circ\text{C} - T_{T8} = 370 \text{ }^\circ\text{C})$	$(H_7 = 451 / H_8 = 443)$	0.52365	0.05	$(H_7 = H_8)$
$H(T_{T8} = 370 \text{ }^\circ\text{C} - T_{T9} = 380 \text{ }^\circ\text{C})$	$(H_8 = 443 / H_9 = 434)$	0.41278	0.05	$(H_8 = H_9)$
$H(T_{T9} = 380 \text{ }^\circ\text{C} - T_{T10} = 390 \text{ }^\circ\text{C})$	$(H_9 = 434 / H_{10} = 421)$	0.32785	0.05	$(H_9 = H_{10})$
$H(T_{T10} = 390 \text{ }^\circ\text{C} - T_{T11} = 400 \text{ }^\circ\text{C})$	$(H_{10} = 421 / H_{11} = 415)$	0.95387	0.05	$(H_{10} = H_{11})$

individual tempering temperatures. All of the first four results of significance were lower than the chosen limit of confidence (0.00092; 0.00038; 0.00004; 0.00001 < 0.05), which indicates that a hypothesis on the equality of the micro hardness among the tempering temperatures chosen was rejected and there were significant differences between the micro hardness chosen.

In Table 4 comparative results of micro hardness significance are given for neighbouring tempering-temperature pairs. Tempering temperatures rose gradually by 10 °C between two neighbouring specimens. Most of the results of significance in Table 4 were higher than the chosen limit of confidence, which means that the differences in the micro hardness between the two neighbouring specimens were insignificant.

5 CONCLUSION

The time delays are defined between the sinusoidal of the magnetizing current and the maximum amplitudes of the BN voltage signals. Statistical analysis was applied to the assessment of the reliability of micro hardness prediction, based on the time delays of the voltage signals. The aim of the research was to produce small differences in micro hardness of the individual specimens and to find out whether there were significant differences. In addition to the assessment of reliability the power of the statistical

relationship between the time delays and micro hardness in the entire tempering-temperature range was assessed.

The statistical analysis provided the following findings:

- The time delays of the captured BN voltage signal is a good estimator of the micro hardness of the 1C55 heat-treated carbon steel, which is indicated by the high values of the calculated correlation coefficients ($r_{t/HV_{0.2}} = 0.989$).
- The temperature differences ranging by $\Delta T_2 = 25 \text{ }^\circ\text{C}$ give us significant differences in micro hardnesses, determined from the time delays, between neighbouring tempering-temperature pairs.
- The temperature differences ranging by $\Delta T_1 = 10 \text{ }^\circ\text{C}$ show mostly insignificant differences in micro hardnesses, determined from time delays, between neighbouring tempering-temperature pairs.
- The results of the micro hardness assessed from the time delay of the BN differ from the average by 16 HV from the measured Vickers hardness.

We estimated that this new micro magnetic method of signal processing is fast, reliable, repeatable and represents a successful estimator for determining the micro hardness of the steel 1C55 in different states.

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